# The Crystal Structure of $\mathrm{Cu}_{4} \mathrm{Bi}_{5} \mathrm{~S}_{10}$ and its Relation to the Structures of Hodrushite and Cuprobismutite 

By K. Mariolacos, V. Kupčík, Masaaki Ohmasa and G. Miehe<br>Mineralogisch-Kristallographisches Institut der Universität Göttingen, D-3400 Göttingen, Germany (BRD)

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#### Abstract

The crystal structure of $\mathrm{Cu}_{4} \mathrm{Bi}_{5} \mathrm{~S}_{10}$ has been determined from three-dimensional intensity data. Two formula units are contained in a monoclinic cell, space group $C 2 / m$, with $a_{0}=17.539$ (3), $b_{0}=3.931$ (13), $c_{0}=12.847$ (3) $\AA$ and $\beta=108.0$ (1). The structure was solved by Patterson, Fourier and difference syntheses, and was then refined by least-squares calculations. The final discrepancy index for all 796 independent reflexions is $12 \cdot 2 \%$. Bismuth atoms have both octahedral and tetragonal pyramidal coordinations, and Cu atoms are surrounded by four S atoms. Two of the cell constants of $\mathrm{Cu}_{4} \mathrm{Bi}_{5} \mathrm{~S}_{10}$ are similar to those of hodrushite, $\mathrm{MBi}_{5} \mathrm{Cu}_{4} \mathrm{~S}_{11}[\mathrm{M}: \mathrm{Bi}, \mathrm{Fe}, \mathrm{Pb} ; \mathrm{Kupči}$, V. \& Makovický, E. (1968). Neues Jb. Miner. Mh. pp. 236-237], and of cuprobismutite [Nuffield, E. W. (1952). Amer. Min. 37, 447-452]. From the relation between the structures of $\mathrm{Cu}_{4} \mathrm{Bi}_{5} \mathrm{~S}_{10}$ and hodrushite, the structure of cuprobismutite was proposed. The coincidence between observed and calculated intensities of the powder data is satisfactory. These three substances are described as layer structures. Both $\mathrm{Cu}_{4} \mathrm{Bi}_{5} \mathrm{~S}_{10}$ and cuprobismutite have singlelayer structures, and hodrushite has a double-layer structure and contains one layer of $\mathrm{Cu}_{4} \mathrm{Bi}_{5} \mathrm{~S}_{10}$ and one layer of cuprobismutite.


## Experimental

In a systematic investigation involving synthetic and crystallographic studies of the $\mathrm{Pb}-\mathrm{Cu}-\mathrm{Bi}-\mathrm{S}$ system, a new compound has been obtained. The material was synthesized by a chemical transport reaction by the method of Rabenau \& Rau (1969). A powder of 3 parts $\mathrm{PbS}, 5$ parts $\mathrm{Bi}_{2} \mathrm{~S}_{3}$ and 1 part $\mathrm{Cu}_{2} \mathrm{~S}$ was mechanically mixed to form the charge for this synthesis. This mixed sulphide was sealed in a silica glass tube together with a transport material $(0 \cdot 1 N \mathrm{HCl}$ solution). The tube. which was tilted at $20^{\circ}$ from the horizontal, was


Fig. 1. Patterson projection $P(u w)$ of $\mathrm{Cu}_{4} \mathrm{Bi}_{5} \mathrm{~S}_{10}$.
kept at $340 \pm 10^{\circ} \mathrm{C}$ with a temperature gradient of $10^{\circ} \mathrm{C}$ for 11 days. In the product, larger crystals of $\mathrm{Bi}_{2} \mathrm{~S}_{3}(\sim 200 \times 200 \times 800 \mu \mathrm{~m})$ with small amounts of Pb and Cu , and fine needles of a new compound ( $\sim 6 \times$ $6 \times 200 \mu \mathrm{~m}$ ) were found. The chemical composition of this new compound was determined by electron microprobe analyses as $\mathrm{Cu}_{4} \mathrm{Bi}_{5} \mathrm{~S}_{10}$.
A crystal of size $7 \times 7 \times 250 \mu \mathrm{~m}$ was selected. Weissenberg photographs were measured and these data were used to refine the cell constants by a least-squares method. These cell constants are similar to those predicted by Kodera, Kupčík \& Makovický (1970). The only observed systematic absence is $h k l$ with $h+k \neq 2 n$, which indicates $C 2$ (No. 5), $C m$ (No. 8) and $C 2 / m$ (No. 12) as possible space groups.

## Crystal data

$\mathrm{Cu}_{4} \mathrm{Bi}_{5} \mathrm{~S}_{10}$, monoclinic, space group C2/m. $a_{0}=$ $17.539 \pm 0.003, b_{0}=3.931 \pm 0.013, c_{0}=12.847 \pm 0.003 \AA$; $\beta=108.0 \pm 0 \cdot 1^{\circ} ; Z=2 ; D_{\text {calc }}=6.39 \mathrm{~g} \mathrm{~cm}^{-3} ; \mu=1154$ $\mathrm{cm}^{-1}(\mathrm{Cu} \mathrm{K} \mathrm{\alpha}$ radiation). The elongation axis is [010].

The intensities were collected from layers $h 0 l$ to $h 3 l$ by an automatic Weissenberg diffractometer (STADI-2 system of Stoe) with Ni -filtered Cu radiation. Because the crystal was very small, diffracted intensities were rather weak. Therefore the intensities of the independent reflexions were measured twice to minimize the error of the measurements. The two sets of data were averaged, and then corrected for the Lorentz-polarization factor. No absorption correction was made, because $\mu r$ was about 0.4 . Of the 796 observed reflexions, 105 had an intensity lower than $2 \sigma(I)$ and were considered as unobserved. For an unobserved reflexion. $\sigma(F) / 3$ was assigned as $F$ observed.

## Structure determination

The reflexions from the layers with $k=2 n(n=0,1)$ show approximately the same intensity distributions. The intensity distributions of the odd layers are similar to each other, but different from those of even layers. This feature suggests that most atoms should be on the planes $y=0$ and $\frac{1}{2}$, and that $C 2$ could be omitted as a possible space group. Cm was also excluded by the interpretation of the Patterson porjection, $P(u w)$ (Fig. 1). The $x$ and $z$ parameters of the heavy atoms (bismuth) were determined from $P(u w)$, and those of the copper and the sulphur atoms were found from the succeeding Fourier and difference syntheses. The $y$ parameters of the atoms, which are either 0 or $\frac{1}{2}$, were determined from generalized projections $P_{1}(u w)$ and $\varrho_{1}(x z)$. The discrepancy index $R$ at this stage was $20 \%$. The positional parameters and isotropic temperature factors of the atoms were refined by the full-matrix least-squares program ORFLS (Busing, Martin \& Levy, 1962). Scattering factors for neutral atoms from International Tables for X-ray Crystallography (1968) were used. Individual weights, $1 / \sigma^{2}\left(F_{0}\right)$, based on counting statistics were used during the refinements. The final discrepancy index $R=\sum\left\{| | F_{o}\left|-\left|F_{c}\right|\right|\right\} / \sum\left|F_{o}\right|$ was $12 \cdot 2 \%$, and the weighted discrepancy index $R^{\prime}=\left\{\sum w\left(\left|F_{o}\right|-\right.\right.$ $\left.\left.\left|F_{c}\right|\right|^{2}\right\}^{1 / 2} /\left\{\sum w\left|F_{o}\right|^{2}\right\}^{1 / 2}$ was $8.3 \%$ for all 796 reflexions, and $10.6 \%$ and $8.1 \%$ respectively when unobserved reflexions were excluded. A three-dimensional difference synthesis showed no significant anomaly at this stage. Refinements of anisotropic temperature factors were not made, because the very low diffracted intensities from the crystal were expected to contain rather high experimental errors.

The atomic positions and the isotropic temperature factors are listed in Table 1. The observed and the calculated structure factors are presented in Table 2.

Table 1. Fractional coordinates and temperature factors Estimated standard deviations in parentheses.

|  | $x$ | $y$ | $z$ | $B$ |
| :---: | :---: | :---: | :---: | :---: |
| $\mathrm{Bi}(1)$ | 0 | 0 | 0 | $1 \cdot 43$ (7) |
| Bi(2) | $0 \cdot 1963$ (2) | 0 | $0 \cdot 3427$ (2) | $2 \cdot 28$ (6) |
| $\mathrm{Bi}(3)$ | $0 \cdot 4127$ (2) | 0 | $0 \cdot 2240$ (2) | $2 \cdot 26$ (6) |
| $\mathrm{Cu}(1)$ | 0.2076 (7) | $\frac{1}{2}$ | 0.0608 (9) | 4-16 (25) |
| $\mathrm{Cu}(2)$ | $0 \cdot 4018$ (6) | $\frac{1}{2}$ | 0.4736 (8) | $3 \cdot 23$ (22) |
| S(1) | 0.0443 (8) | $\frac{1}{2}$ | $0 \cdot 1684$ (11) | $0 \cdot 15$ (25) |
| S(2) | $0 \cdot 2586$ (8) | $\frac{1}{2}$ | $0 \cdot 2452$ (11) | 0.46 (26) |
| S(3) | $0 \cdot 3364$ (8) | 0 | 0.4962 (11) | 0.29 (25) |
| S(4) | $0 \cdot 3463$ (9) | $\frac{1}{2}$ | 0.0268 (12) | $0 \cdot 83$ (28) |
| S(5) | $0 \cdot 4818$ (8) | $\frac{1}{2}$ | $0 \cdot 3613$ (13) | $0 \cdot 42$ (26) |

## Description of the structure

The structure of $\mathrm{Cu}_{4} \mathrm{Bi}_{5} \mathrm{~S}_{10}$ is illustrated in Fig. 2. The interatomic distances are given in Table 3 and the bond angles are given in Table 4.
$\mathrm{Bi}(1)$ is at the origin of the unit cell, and is surrounded by six sulphur atoms in octahedral coordination. An octahedral coordination of bismuth in sulphosalts has also been found in the structures of hodrushite
and $\mathrm{CuBi}_{5} \mathrm{~S}_{8}$ (Ohmasa \& Nowacki, 1973). The other two bismuth atoms, $\operatorname{Bi}(2)$ and $\operatorname{Bi}(3)$, occupy distorted tetragonal pyramids with five sulphur atoms. Of the five distances, three distances $(\sim 2.7 \AA)$ are shorter than the other two ( $\sim 3.0 \AA$ ). This coordination around bismuth is the one typically found in sulphosalt structures (Kupčík, 1972).
$\mathrm{Cu}(1)$ is surrounded by three sulphur atoms in a
Table 2. Observed and calculated structure factors



Fig. 2. The structure of $\mathrm{Cu}_{4} \mathrm{Bi}_{5} \mathrm{~S}_{10}$ projected on (010). The open circles indicate the atoms at $y=0$, and the shaded circles indicate the atoms at $y=\frac{1}{2}$. The smallest circles represent copper atoms, the medium circles and the double circles bismuth atoms, and the largest circles sulphur atoms.
trigonal planar coordination ( $\mathrm{Cu}-\mathrm{S}: 2 \cdot 30 \AA$ ), and the fourth sulphur atom occupies the apex of a trigonal pyramid at a distance of $2 \cdot 60 \AA$. The four sulphur atoms around $\mathrm{Cu}(2)$ form a distorted tetrahedron.

## Structural relation between $\mathrm{Cu}_{4} \mathrm{Bi}_{5} \mathrm{~S}_{10}$, hodrushite and cuprobismutite

The structure of $\mathrm{Cu}_{4} \mathrm{Bi}_{5} \mathrm{~S}_{10}$ is closely related to that of hodrushite, $\mathrm{MCu}_{4} \mathrm{Bi}_{5} \mathrm{~S}_{11}(\mathrm{M}: \mathrm{Bi}, \mathrm{Fe}, \mathrm{Pb} ; \mathrm{Kodera}$ et al., 1970). The Patterson projection, $P(u w)$, of hodrushite shows a very strong maximum at $u=0.189$ and $w=$ $0 \cdot 548$ (Fig. 3). If the origin of a second identical $P(u w)$ projection is shifted to coincide with this maximum in the first $P(u w)$ projection, then the positions and the relative heights of most peaks in both projections coincide. This indicates that the vector between the origin and the maximum ( $u=0.189, w=0.548$ ) is a pseudotranslation vector. This vector divides the unit cell of hodrushite into two parts* where $a_{0}$ and $b_{0}$ of the two

[^0]Table 3. Interatomic distances in $\mathrm{Cu}_{4} \mathrm{Bi}_{5} \mathrm{~S}_{10}$
Estimated standard deviations are in parentheses.

|  | S(1) | S(2) | S(3) | S(4) | S(5) |
| :---: | :---: | :---: | :---: | :---: | :---: |
| Bi(1) | $2 \cdot 846$ (9) (4×) |  |  | 2.821 (20) ( $2 \times$ ) |  |
| $\mathrm{Bi}(2)$ |  | 2.735 (12) (2×) | $\begin{aligned} & 2 \cdot 630(17)(1 \times) \\ & 3 \cdot 035(12)(2 \times) \end{aligned}$ |  |  |
| Bi(3) | $2 \cdot 619$ (19) (1×) |  |  | 3.131 (12) ( $2 \times$ ) | $2 \cdot 666$ (10) ( $2 \times$ ) |
| $\mathrm{Cu}(1)$ |  | $2 \cdot 258$ (18) (1×) |  | $\begin{aligned} & 2 \cdot 316 \text { (11) }(2 \times) \\ & 2 \cdot 602(25)(1 \times) \end{aligned}$ |  |
| $\mathrm{Cu}(2)$ |  |  | 2.339 (12) (2x) |  | $2 \cdot 304$ (21) (1x) |
|  |  |  |  |  | $2 \cdot 446$ (19) (1×) |

Table 4. Bond angles in $\mathrm{Cu}_{4} \mathrm{Bi}_{5} \mathrm{~S}_{10}$
Estimated standard deviations are in parentheses.
Superscript

| none | $x$ | $y$ | $z$ |
| :---: | ---: | ---: | ---: |
| i | $x$ | $-1+y$ | $z$ |
| ii | $\frac{1}{2}-x$ | $\frac{1}{2}+y$ | $1-z$ |
| iii | $\frac{1}{2}-x$ | $-\frac{1}{2}+y$ | $1-z$ |
| iv | $\frac{1}{2}-x$ | $-\frac{1}{2}+y$ | $-z$ |
| v | $\frac{1}{2}-x$ | $-\frac{1}{2}+y$ | $-z$ |
| vi | $\frac{1}{2}+x$ | $-\frac{1}{2}+y$ | $z$ |
| vii | $-\frac{1}{2}+x$ | $-\frac{1}{2}+y$ | $z$ |


| $\mathrm{S}(1)-\mathrm{Bi}(1)-\mathrm{S}\left(1^{\text {i }}\right.$ ) | 87.4 (3) ${ }^{\circ}$ |
| :---: | :---: |
| $\mathrm{S}\left(1^{\text {vi }}\right)-\mathrm{Bi}\left(1^{\text {vi }}\right)-\mathrm{S}\left(1^{v}\right)$ | $92 \cdot 7$ (3) |
| $\mathrm{S}(1)-\mathrm{Bi}(1)-\mathrm{S}\left(4^{v}\right)$ | 87.3 (4) |
| $\mathrm{S}\left(1^{\text {vid }}\right)-\mathrm{Bi}\left(1^{\text {iv }}\right)-\mathrm{S}(4)$ | 92.7 (4) |
| $\mathrm{S}(2)-\mathrm{Bi}(2)-\mathrm{S}\left(2^{\mathbf{i}}\right)$ | 91.9 (4) |
| $\mathrm{S}(2)-\mathrm{Bi}(2)-\mathrm{S}(3)$ | $86 \cdot 0$ (5) |
| $\mathrm{S}(2)-\mathrm{Bi}(2)-\mathrm{S}\left(3^{1 i}\right)$ | $92 \cdot 0$ (3) |
| $\mathrm{S}(2)-\mathrm{Bi}(2)-\mathrm{S}\left(3^{\text {iii }}\right)$ | $164 \cdot 8(1 \cdot 3)$ |
| S(3)-Bi(2)-S(3 ${ }^{\text {ii }}$ ) | $79 \cdot 6$ (4) |
| $S\left(3^{\text {ii }}\right)-\mathrm{Bi}(2)-S\left({ }^{\text {ini }}\right)$ | $80 \cdot 7$ (3) |
| $\mathrm{S}\left(1^{\mathrm{vi}}\right)-\mathrm{Bi}(3)-\mathrm{S}(4)$ | 85.2 (4) |
| $\mathrm{S}\left(1^{\mathrm{vi}}\right)-\mathrm{Bi}(3)-\mathrm{S}(5)$ | $85 \cdot 6$ (5) |
| $\mathrm{S}(4)-\mathrm{Bi}(3)-\mathrm{S}\left(4^{\mathbf{i}}\right)$ | $77 \cdot 8$ (3) |


| $\mathrm{S}(4)-\mathrm{Bi}(3)--\mathrm{S}(5)$ | 92.9 (3) ${ }^{\circ}$ |
| :---: | :---: |
| $\mathrm{S}(4)-\mathrm{Bi}(3)--\mathrm{S}\left(5^{\text {i }}\right.$ ) | $167 \cdot 4$ (1.4) |
| $\mathrm{S}(5)-\mathrm{Bi}(3)-\mathrm{S}\left(5^{\text {i }}\right.$ ) | $95 \cdot 0$ (3) |
| $\mathrm{S}(2)-\mathrm{Cu}(1)-\mathrm{S}(4)$ | $95 \cdot 1$ (7) |
| $\mathrm{S}(2)-\mathrm{Cu}(1)-\mathrm{S}\left(4^{\text {iv }}\right.$ ) | 118.6 (5) |
| $\mathrm{S}(4)-\mathrm{Cu}(1)-\mathrm{S}\left(4^{\text {iv }}\right)$ | $100 \cdot 4$ (6) |
| $\mathrm{S}\left(4^{\text {iv }}\right)-\mathrm{Cu}(1)-\mathrm{S}\left(4^{\text {v }}\right.$ ) | $116 \cdot 2$ (5) |
| $\mathrm{S}(3)-\mathrm{Cu}(2)-\mathrm{S}\left(3^{\mathrm{i}}\right)$ | 114.4 (6) |
| $\mathrm{S}(3)-\mathrm{Cu}(2)-\mathrm{S}(5)$ | 119.7 (6) |
| $\mathrm{S}\left(3^{\text {vii }}\right)-\mathrm{Cu}\left(2^{\text {iid }}\right)-\mathrm{S}\left(5^{\text {iii }}\right)$ | 101.6 (5) |
| $\mathrm{S}\left(5^{\mathrm{vii}}\right)-\mathrm{Cu}\left(2^{\text {iii }}\right)-\mathrm{S}\left(5^{\mathrm{iij}}\right)$ | $92 \cdot 1$ (7) |

parts are common to the original unit cell (Fig. 4). The resultant two sets of $c_{0}$ and $\beta$ coincide with the $c_{0}$ and the $\beta$ of $\mathrm{Cu}_{4} \mathrm{Bi}_{5} \mathrm{~S}_{10}$ and of cuprobismutite (Table 5). Therefore the unit cells of the $\mathrm{Cu}_{4} \mathrm{Bi}_{5} \mathrm{~S}_{10}$ and of cuprobismutite are joined with their $b_{0}$ in antiparallel relationship to form the hodrushite cell. Besides the pseudotranslation, the structure of hodrushite has partial symmetry elements, twofold axes and $2_{1}$ axes in good approximation, as illustrated in Fig. 4. These partial symmetry elements are valid as the symmetry elements in the $\mathrm{Cu}_{4} \mathrm{Bi}_{5} \mathrm{~S}_{10}$ structure, where the arrangement of atoms is practically the same as that in the corresponding part of hodrushite (Fig. 5).

Table 5. The cell constants of $\mathrm{Cu}_{4} \mathrm{Bi}_{5} \mathrm{~S}_{10}$, hodrushite and cuprobismutite

| $\mathrm{Cu}_{4} \mathrm{Bi}_{5} \mathrm{~S}_{10}$ <br> This study |  | Hodrushite | Cuprobismutite |  |
| :---: | :---: | :---: | :---: | :---: |
|  |  | Kodera et al. | This study | Nuffield |
| $a_{0}$ | 17.539 A | $17.575 \AA$ | $17.52 \pm 0.02 \AA$ | $17.65 \AA$ |
| $b_{0}$ | 3.931 | 3.927 | $3.912 \pm 0.001$ | 3.93 |
| $c_{0}$ | $12 \cdot 847$ | 27-205 | $15 \cdot 24 \pm 0.01$ | 15.24 |
| $\beta$ | $108.0^{\circ}$ | $92^{\circ} 9^{\prime}$ | $101 \cdot 26 \pm 0.07^{\circ}$ | $100 \cdot 5^{\circ}$ |

## Proposed structure of cuprobismutite

The considerations in the preceding section enable the structure of cuprobismutite to be predicted. The systematic absence obseived by Nuffield permits $C 2 / m$, $C 2$ and Cm as the possible space groups. Because $C 2 / m$ is common to bismuth sulphosalts and its symmetry coincides with the partial symmetry of hodrushite, the space group of cuprobismutite is assumed to be $C 2 / m$. In order to choose the symmetry centre as the origin of the cuprobismutite cell, its origin must be shifted $\mathrm{c}_{0} / 2$ of the $\mathrm{Cu}_{4} \mathrm{Bi}_{5} \mathrm{~S}_{10}$ cell in the structure of hodrushite (Fig. 4). The proposed structure of cuprobismutite is illustrated in Fig. 5. The chemical formula which corresponds to this structure is $\mathrm{Cu}_{4} \mathrm{Bi}_{7} \mathrm{~S}_{12}$ instead of the formula $\mathrm{CuBiS}_{2}$ proposed by Nuffield. The deviation of the composition from $\mathrm{CuBiS}_{2}$ has recently been suggested by synthetic studies in the system $\mathrm{Bi}_{2} \mathrm{~S}_{3}-\mathrm{Cu}_{2} \mathrm{~S}$ by Buhlmann (1971), and in the system $\mathrm{Cu}-\mathrm{Bi}-\mathrm{S}$ by Sugaki \& Shima (1971). The deviation of the composition from $\mathrm{CuBiS}_{2}$ was also noted by Taylor, Radtke \& Christ (1973) from electron microprobe analyses of natural specimens.

Unfortunately, attempts to synthesize single crystals of this compound have failed. However, a synthetic fine powder was kindly supplied by Professor Moh (Heidelberg), and the validity of the proposed structure could be tested. Table 6 shows the comparison of the observed and the calculated intensities. The $d$ spacings were measured with a Guinier camera (Cu radiation/quartz monochromator). The intensities are the average of the photometric measurements from Guinier photographs and the measurements from a powder diffractometer. Preferred orientation was observed, but was almost eliminated by careful preparation of the specimens. All observed peaks correspond to the re-
flexions with large calculated values of $I$ and the agreement of $I_{\text {calc }}$ with $I_{\text {obs }}$ is satisfactory. The discrepancy index $R^{\prime \prime}=\sum \mid I_{\text {obs }}-I_{\text {cald }} / \sum I_{\text {obs }}$ is $16 \%$ for the reflexions listed in Table 6, which confirms the crystal structure. The positional coordinates of atoms and the isotropic temperature factors used for the calculations of intensities are listed in Table 7. Further attempts to postulate a structure with more copper atoms led to a poorer fit between $I_{\text {obs }}$ and $I_{\text {calc }}$.
The relationships between the structures of $\mathrm{Cu}_{4} \mathrm{Bi}_{5} \mathrm{~S}_{10}$, hodrushite and cuprobismutite are thus clarified. The common features of these three compounds are as follows: (a) they belong to the monoclinic system and their space group is $C 2 / m$ (or $C m$ ), (b) two of the three cell dimensions are 17 and $4 \AA$, (c) the octahedrally coordinated heavy atom occupies the symmetry centre or the origin.
These structures may be described as layer structures; the structures of $\mathrm{Cu}_{4} \mathrm{Bi}_{5} \mathrm{~S}_{10}$ and cuprobismutite


Fig. 3. Patterson projection $P(u w)$ of hodrushite. The $s_{s} c_{0}$, ${ }_{\mathbf{H}} c_{0}$ and ${ }_{\mathrm{CB}_{B} c_{0} \text { correspond to the } c \text { axes of } \mathrm{Cu}_{4} \mathrm{Bi}_{5} \mathrm{~S}_{10} \text {, hodrushite }}$ and cuprobismutite respectively.
are composed of different single fundamental layers and hodrushite has one of each of these fundamental layers. Different combinations of these fundamental layers, for instance two $\mathrm{Cu}_{4} \mathrm{Bi}_{5} \mathrm{~S}_{10}$ layers and one cuprobismutite layer, are also possible, and such substances may be synthesized under appropriate conditions or may be found in nature. This relationship is similar to that found in the structures of the rathite group (Le Bihan, 1962).

Hodrushite and $\mathrm{Cu}_{4} \mathrm{Bi}_{5} \mathrm{~S}_{10}$ have not as yet been found in the synthetic studies of the systems $\mathrm{Cu}-\mathrm{Bi}-\mathrm{S}$ by


Fig. 4. The relationship between the unit cells of hodrushite, $\mathrm{Cu}_{4} \mathrm{Bi}_{5} \mathrm{~S}_{10}$ and cuprobismutite. The smallest cell drawn with the dashed lines indicates the unit cell of $\mathrm{Cu}_{4} \mathrm{Bi}_{5} \mathrm{~S}_{10}$, and the cell of medium size drawn with both dashed and full lines indicates the cuprobismutite cell. The symmetry and the partial symmetry elements of hodrushite are also illustrated together with the symmetry elements of $\mathrm{Cu}_{4} \mathrm{Bi}_{5} \mathrm{~S}_{10}$. A number of the symmetry elements of $\mathrm{Cu}_{4} \mathrm{Bi}_{5} \mathrm{~S}_{10}$ do not fit the structure of hodrushite, because hodrushite consists of two different layers.


Fig. 5. The structure of hodrushite. The part surrounded by the full lines represents the structure of cuprobismutite.

Sugaki \& Shima, and $\mathrm{Bi}_{2} \mathrm{~S}_{3}-\mathrm{Cu}_{2} \mathrm{~S}$ by Buhlmann. This may be attributed to the following reason. Since an

## Table 6. Powder data of cuprobismutite

The first five columns represent the results of present studies. The $d_{\text {obs }}$ were measured by a Guinier camera of a radius 57.3 mm with Cu radiation monochromated by a single crystal of quartz. $I_{\mathrm{calc}}$, obtained after multiplication by the Lp factors for the Debye-Scherrer geometry, is listed for comparison with $I_{\text {obs. }}$. Data by Taylor et al. (1973) are quoted in the next two columns, by Berry \& Thompson (1962) in the centre two columns, and by Nuffield (1952) in the last two columns.

| nk1 | ${ }^{\text {casale }}$ ( ${ }^{(R)}$ | dobs ${ }^{(k)}$ | ${ }^{\text {abs }}$ | ${ }_{\text {cale }}$ | ${ }^{\text {obs }}$ ( ${ }^{(9)}$ | ${ }^{1}$ obs | obs | $\mathrm{I}_{0 \rightarrow s}$ | $\mathrm{dobs}^{(x)}$ | ${ }^{1}$ obs |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| 201 | 6.89 |  |  |  | 6.94 6.28 | 12 35 |  |  |  |  |
| ${ }_{003}^{202}$ | 6.28 4.98 | 4.98 | $\stackrel{20}{8}$ | 16 | 6.28 4.96 | 35 15 | 6.24 | 20 | 6.15 | 20 |
| 801 | 4.36 | 4.36 |  | 3 |  |  |  |  |  |  |
| 400 | 4.29 | 4.30 | 13 | 15 | 4.32, | 40 | 4.31 | 30 | 4.32 | 20 |
| 40. | 3.93 |  |  |  | 3.941 |  |  |  |  |  |
| \%004 | ${ }_{3}^{3.74} 19$ | 3.73 | 5 | 8 |  |  |  |  |  |  |
| ? 04 | 3.70 | 3.70 | 12 | 15 |  |  |  |  |  |  |
| ${ }_{601}^{111}$ | 3.66 3.62 | 3.66 3.62 | 48 | ${ }_{4}^{4}$ | 3.632 | 10 | 3.65 | 408 | 3.64 | 300 |
| ${ }_{402}$ | 3.46) | 3.45 | 28 | 18 | 3.452 | 45 | 3.47 | 10 | 3.46 | 5 |
| 112 | 3.34 | 3.33 | 2 | 2 |  |  |  |  |  |  |
| 311 310 | 3.23 3.23 3.25 | 3.22 | 49 | 55 | 3.220 | 70 | 3.23 | 40 | 3.25 | 40 |
| 204 | 3.21 |  |  |  |  |  |  |  |  |  |
| $\begin{aligned} & 113 \\ & 312 \\ & 311 \end{aligned}$ | $\left.\begin{array}{l} 3.09 \\ 3.09 \\ 3.09 \end{array}\right\}$ | 3.08 | 100 | 100 | 3.090 | 100 | 3.10 | 100 | 3.07 | 100 |
| 005 | 2.99 2.98) | 2.98 | 9 | 11 | 2.971 | 12 | 2.96 | 58 | 2.95 | 50 |
| 601 | 2.921 | 2.919 | 10 | 6 |  |  |  |  |  |  |
| $\begin{aligned} & 602 \\ & 600 \\ & 1313 \\ & 312 \end{aligned}$ | $\begin{aligned} & 2.867 \\ & 2.863 \\ & 2.858 \\ & 2.853 \\ & 2.853 \end{aligned}$ | 2.854 | 29 | 20 | 2.860 | 30 | 2.86 | 108 | 2.86 | 50 |
| $\begin{aligned} & 714 \\ & \hline 03 \\ & 001 \\ & 005 \\ & 005 \end{aligned}$ | $\begin{aligned} & 2.730 \\ & 2.723 \\ & 2.716 \\ & 2.716 \end{aligned}$ | 2.122 | 55 | 58 | 2.721 | 80 | 2.73 | 60 | 2.73 | 60 |
| 114 | 2.614 | 2.613 | 3 | 3 |  |  |  |  |  |  |
| $\begin{aligned} & 314 \\ & 318 \\ & 404 \\ & 404 \end{aligned}$ | $\begin{aligned} & 2.589 \\ & 3.584 \\ & 3.5(5) \\ & 2.581 \end{aligned}$ | 2.585 | 14 | 13 | 2.580 | 40 | 2.58 | 10 | 2.59 | 5 |
| 512 511 515 | 2.559 2. 284 2.48 | 2.555 | 11 9 | 10 6 3 | 2.501 | 10 | 2.49 | 5 | 2.49 | 5 |
| 513 512 | 2.443 2.337 | 2.4.437 | 3 | ${ }_{3}^{3}$ |  |  |  |  |  |  |
| $\begin{aligned} & 315 \\ & 314 \end{aligned}$ | $\begin{aligned} & 2.326 \\ & 2.321 \end{aligned}$ | 2.324 | 3 | 3 |  |  |  |  |  |  |
| $\frac{115}{605}$ | $\begin{aligned} & 2.306 \\ & 2.305 \end{aligned}$ | 2.303 | 6 | 7 |  |  |  |  |  |  |
| $\begin{aligned} & 514 \\ & 206 \end{aligned}$ | $\begin{aligned} & 2.287 \\ & 2.278 \\ & 2.28 \end{aligned}$ | 2.282 | 3 | 4 |  |  |  |  |  |  |
| E01 | 2.187 | 2.188 | 2 | 3 |  |  |  |  |  |  |
| $\begin{gathered} 207 \\ 513 \end{gathered}$ | $\left.\begin{array}{l} 2.175 \\ 2.166 \end{array}\right)$ | 2.172 | 20 | 16 | 2.160 | 35 | 2.17 | 20 | 2.17 | 20 |
| 800 | 2.147 | 2.141 | 3 | 3 |  |  |  |  |  |  |
| $\begin{array}{r} \frac{\mathrm{I} 16}{803} \end{array}$ | $\begin{aligned} & 2.129 \\ & 2.129 \end{aligned}$ | 2.128 | 4 | 5 |  |  |  |  |  |  |
| ${ }_{711} 11$ | 2.114, | 2.101 | 5 | 4 |  |  |  |  |  |  |
| $\begin{aligned} & 712 \\ & 506 \\ & \hline 16 \\ & \hline 004 \\ & 315 \\ & 707 \\ & 180 \end{aligned}$ | 2.094 2.093 2.088 2.085 2.083 2.083 2.082 2.079 | 2.088 | 16 | 19 | 2.091 | 30 | 2.09 | 20 | 2.09 | 20 |
| 116 | $\begin{aligned} & 2.046 \\ & 2.043 \end{aligned}$ | 2.043 | 13 | 8 |  |  |  |  |  |  |
| 514 | 1.992 | 1.992 | 13 | 12 | 1.995 | 25 | 2.00 | 5 | 2.00 | 5 |
| $\begin{aligned} & 714 \\ & 020 \end{aligned}$ | $\left.\begin{array}{l} 1.960 \\ 1.956 \end{array}\right)$ | 1.953 | 28 | 27 | 1.955 | 40 | 1.961 | 30 | 1.957 | 30 |
| $\begin{aligned} & 516 \\ & 021 \end{aligned}$ | $\left.\begin{array}{l} 1.941 \\ 1.939 \end{array}\right)$ | 1,941 | 3 | 7 |  |  |  |  |  |  |
| 112 | 1.923 | 1.922 | 4 | 5 |  |  |  |  |  |  |
| $\begin{aligned} & 607 \\ & 605 \\ & 022 \end{aligned}$ | $\begin{aligned} & 1.899 \\ & \left.\begin{array}{l} 1.892 \\ 1.892 \end{array}\right\} \end{aligned}$ | 1.893 | 6 | 5 |  |  |  |  |  |  |
| 221 117 | 1.882 1.880 | 1.877 | 6 | 6 |  |  | 1.817 | 5 | 1.877 | 5 |
| $\begin{aligned} & 008 \\ & 222 \end{aligned}$ | $\begin{aligned} & 1.869) \\ & 1.867 \end{aligned}$ | 1.869 | 3 | 4 | 1.862 | 12 |  |  |  |  |
| $\begin{aligned} & 803 \\ & 117 \end{aligned}$ | 1.846 1.830 | 1.848 1.831 | ${ }_{6}$ | 3 | 1.825 | 15 | 1.831 | 5 | t.829 | 5 |
| 515 023 | 1.826) | 1.820 | 2 | 4 |  |  |  |  |  |  |
| $\begin{aligned} & 420 \\ & 407 \\ & 517 \end{aligned}$ | $\begin{aligned} & 1.780 \\ & 1.779 \\ & 1.799 \end{aligned}$ | 1.780 | 3 | 4 |  |  |  |  |  |  |
| 912 | 1.7309 | 1.740 | 1 | 3 |  |  |  |  |  |  |
| $\begin{gathered} 7003 \\ 224 \\ 608 \end{gathered}$ | $\begin{aligned} & 1.732 \\ & 1.729 \\ & 1.727 \end{aligned}$ | 1.728 | 3 | 4 |  |  |  |  |  |  |
| $\begin{array}{r} x_{32} \\ 606 \\ 913 \\ 9000 \\ 910 \\ 910 \end{array}$ | 1.721 1.721 1.718 1.718 1.715 | 1.718 | 18 | 11 | 1.122 | 30 | 1.719 | 30 | 1.119 | 30 |
| $\begin{aligned} & 714 \\ & 318 \\ & 222 \\ & 317 \end{aligned}$ | $\begin{aligned} & \begin{array}{l} 1.711 \\ 1.701 \\ 1.701 \\ 1.698 \end{array} \end{aligned}$ | 1.701 | 6 | 5 |  |  |  |  |  |  |
| ${ }_{807} 8$ | ${ }^{1.691}$ ) | 1.690 | 8 | 7 |  |  | B broas |  | - |  |

Table 7. Atom parameters of cuprobismutite used to calculate the intensities listed in Table 6

|  | $x$ | $y$ | $z$ | $B$ |
| :--- | :--- | :--- | :--- | :---: |
| $\mathrm{Bi}(1)$ | 0 | 0 | $\frac{1}{2}$ | $2 \cdot 5$ |
| $\mathrm{Bi}(2)$ | $0 \cdot 256$ | $\frac{1}{2}$ | $0 \cdot 129$ | $2 \cdot 5$ |
| $\mathrm{Bi}(3)$ | $0 \cdot 009$ | 0 | $0 \cdot 220$ | $2 \cdot 5$ |
| $\mathrm{Bi}(4)$ | $0 \cdot 174$ | $\frac{1}{2}$ | $0 \cdot 406$ | $2 \cdot 5$ |
| $\mathrm{Cu}(1)$ | $0 \cdot 412$ | 0 | $0 \cdot 021$ | $3 \cdot 5$ |
| $\mathrm{Cu}(2)$ | $0 \cdot 357$ | 0 | $0 \cdot 358$ | $3 \cdot 5$ |
| $\mathrm{~S}(1)$ | $0 \cdot 337$ | $\frac{1}{2}$ | $0 \cdot 006$ | $1 \cdot 5$ |
| $\mathrm{~S}(2)$ | $0 \cdot 033$ | $\frac{1}{2}$ | $0 \cdot 110$ | $1 \cdot 5$ |
| $\mathrm{~S}(3)$ | $0 \cdot 350$ | 0 | $0 \cdot 209$ | $1 \cdot 5$ |
| $\mathrm{~S}(4)$ | $0 \cdot 163$ | 0 | $0 \cdot 275$ | $1 \cdot 5$ |
| $\mathrm{~S}(5)$ | $0 \cdot 017$ | $\frac{1}{2}$ | $0 \cdot 380$ | $1 \cdot 5$ |
| $\mathrm{~S}(6)$ | 0.338 | $\frac{1}{2}$ | $0 \cdot 440$ | $1 \cdot 5$ |

octahedral coordination may be less favourable to bismuth than the other types such as trigonal or tetragonal pyramids, it is likely that other elements (e.g. $\mathrm{Pb}, \mathrm{Fe}$ and so forth), even in small amounts, catalyse the formation of these compounds (Moh, 1973). Therefore $\mathrm{Cu}_{4} \mathrm{Bi}_{5} \mathrm{~S}_{10}$, hodrushite and cuprobismutite may belong to a quarternary system, $\mathrm{M}-\mathrm{Cu}-\mathrm{Bi}-\mathrm{S}(\mathrm{M}: \mathrm{Pb}, \mathrm{Fe})$, with a small amount of M , or they may be formed in the $\mathrm{Cu}-\mathrm{Bi}-\mathrm{S}$ system as a metastable phase only under the condition that a small amount of M is present. A similar behaviour of minor elements has been reported by Bernhardt (1972) in his study on the sulphide system, $\mathrm{Cu}_{2} \mathrm{FeSnS}_{4}-\mathrm{CuFeS}_{2}$.

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# The Structure of 2,3-Dichloro-1,1-dicyano-3,3-diphenylpropene 

By R. Schlodder and James A. Ibers<br>Department of Chemistry, Northwestern University, Evanston, Illinois 60201, U.S.A.

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The crystal and molecular structure of 2,3-dichloro-1,1-dicyano-3,3-diphenylpropene, $\mathrm{C}_{17} \mathrm{Cl}_{2} \mathrm{H}_{10} \mathrm{~N}_{2}$, has been determined from 2067 X-ray data above background collected by counter methods. The material crystallizes with eight molecules in space group Pbca $\left(D_{2 h}^{15}\right)$ of the orthorhombic system in a cell of dimensions $a=15 \cdot 403$ (3), $b=19 \cdot 219$ (4), $c=10 \cdot 138$ (2) $\AA$. The observed and calculated densities are $1 \cdot 39$ (1) and $1.386 \mathrm{~g} \mathrm{~cm}^{-3}$. The structure has been solved by direct methods and refined by full-matrix least-squares methods to a final $R$ index on $F$ of 0.036 . The $\mathrm{C}=\mathrm{C}(\mathrm{CN})_{2}$ portion of the molecule closely resembles that in tetracyanoethylene. Whereas the olefinic $\mathrm{C}-\mathrm{Cl}$ bond length is $1.715(2) \AA$, the paraffinic $\mathrm{C}-\mathrm{Cl}$ bond length is 1.836 (2) $\AA$. The material was prepared by the reaction of diphenylcarbene with 1,1-dichloro-2,2-dicyanoethylene. The structure found here indicates that the reaction involves the insertion of diphenylcarbene into a $\mathrm{C}-\mathrm{Cl}$ bond.

## Introduction

In an earlier investigation of the bonding of the unsymmetrical olefin 1,1-dichloro-2,2-dicyanoethylene to
a transition metal the complex $\operatorname{Pt}\left[\mathrm{Cl}_{2} \mathrm{C}=\mathrm{C}(\mathrm{CN})_{2}\right]$ $\left[\mathrm{P}\left(\mathrm{C}_{6} \mathrm{H}_{5}\right)_{3}\right]_{2}$ was prepared and its structure determined (McAdam, Francis \& Ibers, 1970). Because of the difference in $\mathrm{Pt}-\mathrm{C}$ (of Cl ) versus $\mathrm{Pt}-\mathrm{C}$ (of CN ) bond


[^0]:    * A different choice of axes for hodrushite is used in this paper from that in the papers of Kupčik \& Makovický (1968) and Kodera et al. (1970) to conform to International Tables for $X$-ray Crystallography and to use the same axes for all three substances.

